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# INSTITUT TEHNIČKIH NAUKA SANU

Knez Mihailova 35/IV

11000 Beograd, Srbija

Predmet: Zahtev za pokretanje postupka za izbor Adriane Peleš, dipl. fizičara, istraživača pripravnika u zvanje istraživača saradnika

# NAUČNOM VEĆU INSTITUTA TEHNIČKIH NAUKA SANU

Molim Vas da, u skladu sa Pravilnikom o postupku i načinu vrednovanja, i kvantitativnom iskazivanju naučnoistraživačkih rezultata istraživača (Sl. Glasnik RS, br.38/08), i Pravilnikom o sticanju zvanja istraživača saradnika, naučno veće Instituta tehničkih nauka SANU pokrene postupak za moj izbor u zvanje istraživač saradnik.

Za članove komisije za pripremu izveštaja Naučnom Veću predlažem:

- prof. dr Vladimira Pavlovića, naučnog savetnika Instituta tehničkih nauka SANU
- dr Ninu Obradović, višeg naučnog saradnika Instituta tehničkih nauka SANU
- dr Lidiju Mančić, višeg naučnog saradnika Instituta tehničkih nauka SANU

## U prilogu dostavljam:

- 1. biografiju
- 2. bibliografiju sa kopijom rada
- 3. dokaz o ukupnoj prosečnoj oceni na osnovnim studijama
- 4. dokaz o upisanim doktorskim studijama

U Beogradu, 19. avgust 2013.

Podnosilac zahteva

Adriana Peleš, dipl fizičar

Aguaya Tenen

Istraživač pripravnik ITN SANU

## Biografija Adriane Peleš

Adriana Peleš rođena je 01. 08. 1984. godine u Beogradu. Diplomirala je na Fizičkom fakultetu Univerziteta u Beogradu sa prosečnom ocenom 8,2 i 10 na diplomskom radu "Korišćenje digitalne video kamere u spektroskopiji-kalibracija i primena". Postdiplomske doktorske studije je upisala 2011. godine na matičnom fakuletu, smer primenjena i kompjuterska fizika. Trenutno je na drugoj godini studija.

U Institutu tehničkih nauka SANU je zaposlena od 2012. godine kao istraživač pripravnik. Angažovana je na projektu osnovnih istraživanja OI172057 "Usmerena sinteza, struktura i svojstva multifunkcionalnih materijala" (rukovodilac projekta prof. dr Vladimir Pavlović).

Oblasti interesovanja su joj tehnologija prahova, nano materijali, karakterizacija materijala, polimerni materijali, matematičke analize, dielektirčna svojstva keramika, mehanička aktivacija, numeričke metode.

## Bibliografija Adriane Peleš

## M 23 Međunardoni časopis

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## Učešće na međunarodnim konferencijama i seminarima

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# Influence of Prolonged Sintering Time on Density and Electrical Properties of Isothermally Sintered Cordierite-based Ceramics

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### Abstract:

Mechanical activation is a commonly used and relatively fast and inexpensive procedure for sample preparation before the sintering process. Cordierite, a stoichiometric mixture of three different oxides (2MgO·2Al<sub>2</sub>O<sub>3</sub>·SSiO<sub>2</sub>) is a very attractive, widely used high-temperature ceramic material. The mechanical activation of the starting mixtures with 5.00 mass% TiO<sub>2</sub> was performed in a high energy ball mill during 10-80 min. The applied compaction pressure before the sintering process was 2t/cm², based on our recent investigation. The sintering process was performed at 1350°C for 2h and 4h in air atmosphere. X-ray diffraction was used to analyze the phase composition of non-activated and 80 min activated samples, sintered for 2 and 4h, respectively. Scanning electron microscopy was performed to analyze the microstructure of both compacted and sintered samples. Atomic force microscope was used to investigate the surface of the sintered samples. This paper investigates the influence of prolonged sintering time on the densities of the sintered samples, along with electrical properties.

**Keywords**: Mechanical activation, Density, XRD, SEM, Electrical properties, AFM, Cordierite.

### 1. Introduction

Cordierite  $(2\text{MgO}\cdot2\text{Al}_2\text{O}_3\cdot5\text{SiO}_2)$  based ceramics are widely used in various fields, from substrates for micro-electronic packaging industry to cookware, heat exchangers, glazes for floor tiles. Owing to the very low temperature thermal expansion coeficient  $(20\cdot10^{-7/\circ}\text{C})$  and low relative dielectric constant (~5), these ceramics are also well known by their good thermo-mechanical, chemical and dielectric properties [1,2]. They can be applicable as materials that are exposed to sudden temperature changes [3-7] and also as a semiconducting bearers [8,9]. The temperature range of cordierite sintering is very narrow (1300-1400°C).

Our previous investigations showed a significant influence of mechanical activation, as well as compaction pressure on starting mixtures (kaolin, quarz, magnesium oxide) on lowering of sintering temperature [10]. Mechanically activated components increased energy

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due to induced crystal defects compared to non-activated ones. During mechanochemical treatment, several processes occur: attrition of starting material, crystal lattice destruction, various defects formation, etc. All mentioned processes increase the chance that during heating, processes could be observed at lower temperatures than usual [11]. Furthermore, mechanical activation could affect the final electrical characteristics, so it is very important to approach and understand changes that get introduced into the system while milling.

In this paper, the authors used some starting conditions based on the previous investigation and tried to study the influence of sintering time on density, phase composition as well as on the microstructure along with electrical properties of sintered samples.

### 2. Experimental procedure

Mixtures of Mg(OH)<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, SiO<sub>2</sub> and TiO<sub>2</sub> (all p.a. purity) were used in these experiments. The cordierite ceramics starting mixture, MgO+Al<sub>2</sub>O<sub>3</sub>+SiO<sub>2</sub> in the 2:2:5 ratio, with the addition of 5.00 mass% TiO<sub>2</sub>, were mechanically activated by grinding in a high-energy planetary ball mill.  $ZrO_2$  vessels and balls were used with the powder to balls mass ratio of 1:40. The milling process was performed in air atmosphere for 10, 20, 40 and 80 minutes. The samples were denoted as K-0, K-10, K-20, K-40 and K-80, according to the milling time

The X-ray powder diffraction patterns after milling and sintering were obtained using a Philips PW-1050 diffractometer with  $\lambda \text{Cu-K}_{\alpha}$  radiation and a step/time scan mode of 0.05°/1s. The morphology of the obtained powders and sintered samples were characterized by scanning electron microscopy (JEOL JSM-6390 LV). The powders were crushed and covered with gold in order to perform these measurements.

The pressure used in our experiments was 2t/cm² (approximately 200 MPa). The pressure was performed in a double-sided tool 6 mm in diameter (Hydraulic press RING 14, VEB THURINGER). The density of the specimens was calculated from measurements of their diameter, thickness, and mass. The theoretical density (TD) of mixture is 3.078 g/cm³ and is calculated based on the following equation:

$$\rho_{mixture} = \frac{m_{mixture}}{V_1 + V_2 + V_3 + V_4} \tag{1}$$

where: m – mass of the mixture (30.00 g),  $V_1$ ,  $V_2$ ,  $V_3$  and  $V_4$  – volumes of component  $1-Mg(OH)_2$ , component  $2-SiO_2$ , component  $3-Al_2O_3$  and component  $4-TiO_2$  in the mixture, calculated by usage of TD of each component.

The compacts were sintered isothermally at 1350°C, in air atmosphere for 2h and 4h, with heating rate of 10°C/min in a Lenton furnace.

The topography and roughness of samples were characterized using an atomic force microscope (AFM; Veeco Instruments, model Dimension V). Micrographs were obtained in tapping mode under ambient conditions, using TAP300 tips (resonant frequency 300 kHz, force constant 40 N/m). Roughness data were obtained using diNanoScope software (version 7.0). Scan sizes were 10x10 microns.

### 3. Results and discussion

Mechanically activated samples of cordierite with the addition of 5mass%  $\rm TiO_2$  were sintered for 2 and 4 hours at 1350°C. The density changes of the samples before and after sintering process were investigated. Density values of samples before and after sintering are shown in Tab. I for samples sintered for 2 hours, and in Tab. II for samples sintered for 4 hours. The results shown in Tabs I and II are consistent with the expected results. The density

of the samples was higher than the density of the starting material, also the density increased with prolonged sintering time.

**Tab. I** Mass and densities of green bodies and 2h sintered cordierite samples with addition of 5mass% TiO<sub>2</sub>.

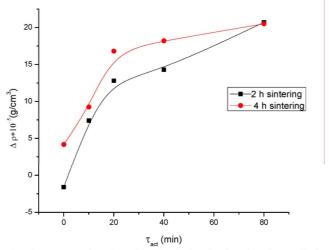
activation time				_
(min)	*m <sub>0</sub> (g)	$\mathbf{m}_{s}(\mathbf{g})$	**ρ₀(g/cm³)	$\rho_{\rm s}({\rm g/cm}^3)$
0	0.2980	0.2845	2.001	1.990
10	0.2991	0.2845	2.070	2.145
20	0.2978	0.2839	2.048	2.178
40	0.2986	0.2845	2.039	2.184
80	0.2982	0.2856	1.962	2.168

**Tab. II** Mass and densities of green bodies and 4h sintered cordierite samples with addition of 5mass% TiO<sub>2</sub>.

 samples with addition of Smass v 110 g.										
activation time			_	_						
(min)	$\mathbf{m}_{0}(\mathbf{g})$	$m_s(g)$	ρ <sub>0</sub> (g/cm <sup>3</sup> )	$\rho_{\rm s}({\rm g/cm}^3)$						
0	0.2980	0.2840	1.975	2.011						
10	0.2978	0.2835	2.072	2.160						
20	0.2980	0.2843	2.035	2.204						
40	0.2982	0.2842	2.050	2.202						
80	0.2983	0.2853	1.968	2.178						

<sup>\*</sup>m<sub>o</sub>, m<sub>s</sub> mass of sample before and after sintering process

In order to compare the results of changes in the density of samples before and after the sintering process, density variation  $\Delta \rho$ , ( $\Delta \rho = \rho_s - \rho_o$ ), as a function of mechanical activation time for both sintering time, 2 and 4 hours, was analyzed. The results are presented at Fig. 1.



**Fig. 1**. Density changes as a function of mechanical activation time for cordierite samples sintered for 2h and 4h at 1350°C.

<sup>\*\*</sup> $\rho_{o}$ ,  $\rho_{s-}$  densities of sample before and after sintering process

The curves show that  $\Delta \rho$  of samples sintered for 2h and 4h increased rapidly in the first 20 minutes of mechanical activation, whereas the densities of samples sintered for 4h have a higher value. We expected these results if we have in mind the great energy that is introduced within the powder during milling process and the starting conditions (balls to powder mass ratio 40:1).

In the first 20 min of mechanical activation, the major changes are taking place in the starting powder material. It causes significant fragmentation and homogenization of the starting ingredients. Thus the activated samples during the sintering process show faster chemical reaction in the solid state and the density curve of the sintered samples as a function of activation time shows that the length of time of mechanical activation significantly increases the density of the obtained material. Samples sintered for 4 hours have a higher change in density than samples sintered for 2 hours. Samples sintered 2 and 4 hours exhibit changes in density sharply in the first 20 minutes of mechanical activation. After 20 min of activation time, density changes are still moving to higher values, but slower.

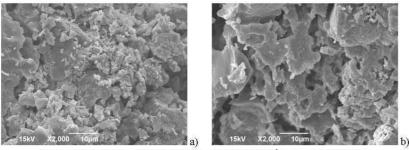


Fig. 2. SEM of non-activated samples sintered at  $1350^{\circ}\mathrm{C}$  for a) 2h and b) 4h.

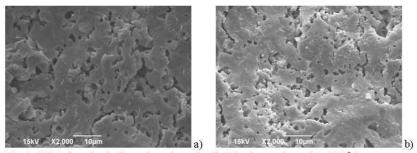


Fig. 3. SEM of mechanically activated samples for 80 minutes sintered at 1350°C for a) 2h and b) 4h.

Figs. 2. and 3. represent scanning electron micrographs of sintered powders. Non-activated powder sintered for 2 hours (Fig. 2. a) shows fine-grained structure, defined grain sintered material, and high porosity between grains. The surface is rough. The particles of the starting material were homogenized, but the extension of time of sintering (Fig. 2. b) at a temperature of 1350°C leads to further convergence of grain, and the reaction sintering process allows better contact of the starting particles. This leads to the formation of larger agglomerates. The porosity of the obtained product is lower than the previous sample sintered for 2h, but still very high. Defined grains are much less conspicuous. Blocks of agglomerates

are formed, due to a longer sintering time. Samples which were mechanically activated for 80 minutes showed a significantly smoother texture.

Porosity was significantly smaller, and was practically a negligible difference in the appearance between the surface of samples sintered for 2 and 4 hours (Fig. 3. a) and b)). The defined grain of the starting powder material flow is not visible. Based on the SEM images, we can see a significant influence of mechanical activation on the sintering process, and the impact of the length of the sintering process on the cordierite sample set.

XRD patterns of samples non-activated and activated for 80 min and sintered for 2h and 4h at 1350°C are shown in Figs. 4. and 5. All obtained intensities were identified by JCPDS cards (084-1220 for cordierite, 083-2241 for TiO<sub>2</sub> rutile, 074-0201 for SiO<sub>2</sub>, 089-7717 for alumina and 084-0377 for magnesium aluminum oxide).

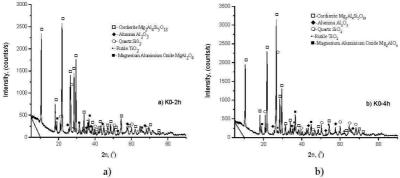


Fig. 4. XRD patterns of non-activated samples sintered at 1350°C for a) 2h and b) 4h.

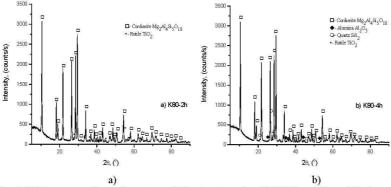


Fig. 5. XRD patterns of samples activated 80 min sintered at 1350°C for a) 2h and b) 4h.

XRD analysis of non-activated samples sintered 2 and 4 hours shows a presence of cordierite formation as well as the remains of the starting oxides. Corund ( $Al_2O_3$ ), quarts ( $SiO_2$ ) and  $MgAl_2O_4$  and  $TiO_2$  rutile were detected. Comparing the diffraction patterns of non-activated samples, sintered 2 and 4 hours, it can be seen that the sample sintered 4 h have higher amounts of cordierite (66% compared with 60% cordierite sintered for 2 hours). The remains of the starting oxides are reduced with the prolonged sintering of the starting material.

Diffraction patterns which are mechanically activated 80 minutes (Fig. 5.) had significantly greater responses compared to non-activated sintered samples. The sample which was sintered for 2 hours showed practically insignificant traces of the starting oxides corundum (2.4%) and quartz (0.9%) and (95.4%) cordierite phase. The sample sintered for 4 hours has no remaining traces of the starting oxides (except rutile TiO<sub>2</sub> content of 2% additives whose presence was expected). Cordierite is present in an amount of 98%, which can be considered a fully completed reaction during the sintering process.

**Tab. III** Electrical properties of cordierite after sintering at  $1350^{\circ}$ C for a) 2h and b) 4 h as a function of milling time (0 - 80 min).

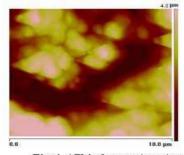
-01	 	

2h	C	ετ
0	0.71	3.7418
10	0.88	4.6468
20	0.88	4.6734
40	0.80	4.3363
80	0.83	4.6462

4h	C	ετ		
0	0.77	4.0973		
10	0.82	4.3248		
20	0.84	4.5167		
40	0.92	4.9741		
80	0.82	4.5992		

Tab. III presents the results of measured capacitance. Electrical properties of the obtained materials were monitored as a function of time of mechanical activation and prolonged sintering time. From these data the value of the dielectric constant was calculated. Electrical measurements were in great accordance with results given previously in the paper regarding XRD patterns. It is obvious that the milling and sintering time have influence on the capacitance and dielectric constant values. Two areas are visible: the first one for samples activated 0 - 20 minutes, and the second for samples activated 20 - 80 min. Both the capacitance and dielectric constant rapidly increase with the first 20 minutes of activation, whereby more significant changes in the sample sintered for 2 hours. After 20 minutes of mechanical activation changes in capacitance and dielectric constant are smaller. Data value for the dielectric constant of the sintered samples for 2 and 4 hours range from 3.74 to 4.97, as expected.

We investigated the influence of sintering time and the influence of activation time by AFM. Due to pictures we made by AFM we can see that roughness of our samples decrease with sintering time and activation time. On the Fig. 6. a) we can see the non-activated sample sintered for 2h (Image roughness is 1039 nm) and has greater roughness than the sample sintered at 1350° for b) 4h (Image roughness is 763 nm). On the Fig. 7. we can see sample activated for 80 minutes and sintered for a) 2h and b) 4h, and the same influence of activation and sintering time are noticed. Image roughness for sample sintered for 2h is 611 nm, and for the sample sintered for 4h is 352 nm.



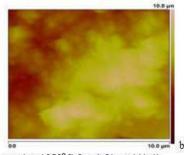


Fig. 6. AFM of non-activated samples sintered at 1350°C for a) 2h and b) 4h.

2)

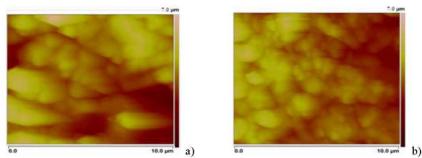


Fig. 7. AFM of 80 minutes activated samples and sintered at 1350°C for a) 2h and b) 4h.

#### 4. Conclusions

Investigation results shows that sintering time, as well as mechanical activation, has influence on the final characteristics of the sintered material. The results that justify prolonged sintering time are:

- Longer sintering time increases the density variation of the sintered samples. The time of mechanical activation has also influence on the density changing of the samples, and increases with increasing sintering time.
- At the same sintering temperature, samples activated 80 minutes have a higher percentage of reacted starting material compared to samples that have not been activated. For samples that were sintered 4 hours and 80 minutes activated starting components were not found. Cordierite ceramics was obtained in an amount of 98%. The remaining 2% is TiO<sub>2</sub> rutile, which was added to the starting mixture as an additive.
- SEM results shows that samples which were mechanically activated for 80 minutes showed a significantly smoother texture. Porosity was significantly smaller than non-activated samples surface, and was practically a negligible difference in the appearance between the surface of samples sintered for 2 and 4 hours.
- Electrical investigation shows the influence of mechanical activation time and sintering time on values of capacitance and dielectric constant. Both capacitance and dielectric constant rapidly increase with first 20 minutes of mechanical activation. Data value for the dielectric constant of the sintered samples 2 and 4 hours range from 3.74 to 4.97, which is in accordance with literature data.
- AFM results shows that samples which were mechanically activated for 80 minutes showed a significantly smoother texture. Image roughness is in decrease with mechanical activation time as well as sintering time.

### Acknowledgement

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 $\it Cadpжaj$  Механичка активација је брза и јефтина процедура која се обично користи за припремање узорака пре процеса синтеровања. Кордијерит, стехиометријска мешавина три различита оксида  $(2MgO\cdot 2Al_2O_3\cdot 5SiO_2)$  је веома атрактиван, и ишроко коришћен високотемпературни керамички материјал. Почетна смеша са 5.00 масених %  $\it TiO_2$  је механички активирана у високоенергетском млину са куглицама у временском интервалу од 10- 80 минута. Пре процеса синтеровања узорци су испресовани под притиском од  $\it 2t/cm^2$ , базирано на нашем пређашњем истраживању. Узорци су синтеровани на температури од  $\it 1350^\circ C$  у периоду од  $\it 2$  и  $\it 4$  сата у атмосфери ваздуха. Дифракцијом  $\it X$  зрака је установљена фаза неактивираних и узорака активираних  $\it 80$  минута, синтерованих  $\it 2$  и  $\it 4$  сата, респективно. Скенирајућом електронском микроскотијом је анализирана микроструктура оба синтерована узорка. Микроскопом атомских сила смо анализирали површину оба узорка. Овај рад иститује утицај продуженог времена синтеровања на густину синтерованих узорака, као и електричне особине.

**Кьучне речи**: Механичка активација, Густина, XRD, SEM, Електичне особине, AFM, Кордијерит.

## FIZIČKI FAKULTET UNIVERZITETA U BEOGRADU

Broj 21202011 Beograd, 27. 12. 2011. godine

Na osnovu člana 161. Zakona o opštem upravnom postupku i člana 4. Pravilnika o sadržaju i obliku obrazaca javnih isprava koje izdaju više škole, fakulteti i univerziteti, po zahtevu, Peleš Adriane izdaje se sledeće

### UVERENJE

PELEŠ (PREDRAG) ADRIANA, rođena 01. 08. 1984. godine u Beogradu, R.Srbija, upisan školske 2003/2004. godine na Studijsku grupu **FIZIKA**, smer: **Primenjena fizika i informatika**, položila je ispite predviđene nastavnim planom i programom navedene Studijske grupe i diplomirala na Fizičkom fakultetu Univerziteta u Beogradu 23.decembra 2011. godine, sa srednjom ocenom 8,20 (osam i 20/100) u toku studija i ocenom 10 (deset i 00/100) na diplomskom ispitu i time stekala visoku stručnu spremu (VII stepen stručne spreme) i stručni naziv

## DIPLOMIRANI FIZIČAR ZA PRIMENJENU FIZIKU I INFORMATIKU

Uverenje se izdaje na lični zahtev, a služi kao dokaz o završenoj visokoj stručnoj spremi (VII stepen stručne spreme) do izdavanja diplome.

Uverenje je oslobođeno plaćanja takse.

D E K A N FIZIČKOG FAKULTETA

Prof. dr Ljubiša Zeković

Naziv proizvoda: UVERENJE O STUDIRANJU A5/OFSET  Proizvodat: OPTIMUM d.o.o. Srbija	Уверење је ослобођено таксе према чл. 18 ставу 1. тач. 6, 7, и 8 Закона о административним таксама ("Службени гласник РС", бр. 49/92, 70/92, 37/93, 44/93, 67/93, 28/94, 30/94, 53/95, 39/96, 42/98).	Ово се уверење може употребити за регулисање војне обавезе, права на дечији додатак, породичне пензије инвалидског додатка, добијања здравствене књижице и легитимације за повлашћену вожњу.	Према статуту факултета - више школе студије трају	dysuca	факултета - више школе у Бърграду	годину студија, школске 2012 2013 године, као ЭСТОВАН	у ТЕОГРАЛУ Република, односно држава Среги Д	NETIEW PREADOR ANDVAHA (Ripesime, ume jedhor od родительа и име)	УВЕРЕЊЕ	11.01. 2013-године	BPOJ D-35/2011	REOFPAN (COLUMNTO)	(назив факултета)- више школе)	РЕПУБЛИКА СРБИЈА
Овлашћене Тице Факилтета - (више школе	, 7, и 8 Закона о административним таксама 30/94, 53/95, 39/96, 42/98).	у́авезе, права на дечији додатак, породичне пензије, в за повлашћену вожњу.	П ГОД МЪ (година-семестара)		DAMERENA N KOLDINETOLA	BAGH) A STYDEHT DAYA COS	BYA, ynucao-na je (Apyty	рођен-а 01.08.1984	Maria	DETOPO TO				